



# First principles study of optical properties of molybdenum disulfide: From bulk to monolayer

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## ARTICLE INFO

### Article history:

Received 19 December 2017

Received in revised form 12 January 2018

Accepted 12 January 2018

Available online 19 January 2018

### Keywords:

Ab-initio

MoS<sub>2</sub>

Optical properties

## ABSTRACT

In this paper, we theoretically study the optical properties of both bulk and monolayer MoS<sub>2</sub> using first-principles calculations. The optical characters such as: dielectric function, optical reflectivity, and electron energy-loss spectrum of MoS<sub>2</sub> are observed in the energy region from 0 to 15 eV. At equilibrium state the dielectric constant in the parallel  $E_{\parallel x}$  and perpendicular  $E_{\parallel z}$  directions are of 15.01 and 8.92 for bulk while they are 4.95 and 2.92 for monolayer MoS<sub>2</sub>, respectively. In the case of bulk MoS<sub>2</sub>, the obtained computational results for both real and imaginary parts of the dielectric constant are in good agreement with the previous experimental data. In the energy range from 0 to 6 eV, the dielectric functions have highly anisotropic, whereas they become isotropic when the energy is larger than 7 eV. For the adsorption spectra and optical reflectivity, both the collective plasmon resonance and  $(\pi + \sigma)$  electron plasmon peaks are observed, in which the transition in  $E_{\parallel x}$  direction is accordant with the experiment data more than the transition in  $E_{\parallel z}$  direction is. The refractive index, extinction index, and electron energy-loss spectrum are also investigated. The observed prominent peak at 23.1 eV in the energy-loss spectra is in good agreement with experiment value. Our results may provide a useful potential application for the MoS<sub>2</sub> structures in electronic and optoelectronic devices.

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## 1. Introduction

Graphene [1], a two-dimensional (2D) materials in the single atomic layer, has attracted great interest for applications in nanoelectronic technology due to their unique transport properties [2–6]. However, the clean graphene does not have a band gap in its electronic structure, which is deemed as a considerable drawback for many applications, in optics and transistor technology. This limitation of graphene leads researchers to investigate alternate materials that are similar to graphene and distinct in their electronic properties. In recent year, transition metal dichalcogenides (TMDs) have become of great

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interest due to their novel mechanical, transport, and optical properties [7–14]. Among them, molybdenum disulfide (MoS<sub>2</sub>) is currently one of most interesting materials for applications in nano-electronic and optoelectronic devices due to its unique electronic, mechanical, transport, and optical properties [15–20].

The MoS<sub>2</sub> bulk is a semiconductor with an indirect electronic band gap of the 1.24 eV opening between the highest energy state of the valence bands located at the  $\Gamma$  point and the lowest energy state of the conduction bands located between the  $\Gamma$  and  $K$  points. Whereas, the MoS<sub>2</sub> monolayer is the semiconductor with a direct band gap of the 1.80 eV [7] opening between the highest/lowest energy state of the valence/conduction bands located at the  $K$  point. Moreover, MoS<sub>2</sub> monolayer has been successfully synthesized by various experimental techniques, such as chemical vapor deposition [21], micromechanical cleavage [22], or liquid exfoliation [23]. The indirect–direct band gap transition makes MoS<sub>2</sub> nanomaterial being a promising potential material for application in optoelectronics [24]. Recently, the structure of both bulk and monolayer MoS<sub>2</sub> has been studied by density functional theory (DFT) [25]. Effect of uniaxial strain and external electric field on electronic properties of both bulk and monolayer MoS<sub>2</sub> has also been considered by different methods [16,26–29]. Besides, optical properties of MoS<sub>2</sub> have attracted many scientists [30–33]. Using DFT calculations, they have shown that the blue-shift phenomenon was observed in O–doped MoS<sub>2</sub> systems [11]. The static dielectric constant of the monolayer MoS<sub>2</sub> with neutral S and Mo vacancies is respectively 1.50 and 2.45, and the S vacancy does not effect on the static dielectric constant of the monolayer MoS<sub>2</sub> [34]. Recently, we theoretically studied the magneto-optical absorption coefficient of MoS<sub>2</sub> on polar substrates in the presence of an external magnetic field [35].

In the present work, we study the optical properties of the MoS<sub>2</sub> using DFT calculations. We calculate the optical characters of both bulk and monolayer MoS<sub>2</sub>. Effect of direction of an external electric field on the dielectric constant and other optical characters is also considered and discussed.

## 2. Theoretical model and method

In this paper, DFT calculations of electronic and optical properties of bulk and monolayer MoS<sub>2</sub> were performed employing the APW + lo method [36]. The full potential which not depends on any predefined form, such as muffin-tin one was adopted in the present calculations. The muffin-tin (MT) sphere radii of atoms constituting the MoS<sub>2</sub> system were supposed to be equal to 2.27 a.u. for Mo and 1.95 a.u. for S in bulk MoS<sub>2</sub> and 2.44 a.u. for Mo and 2.1 a.u. for S in monolayer MoS<sub>2</sub>.

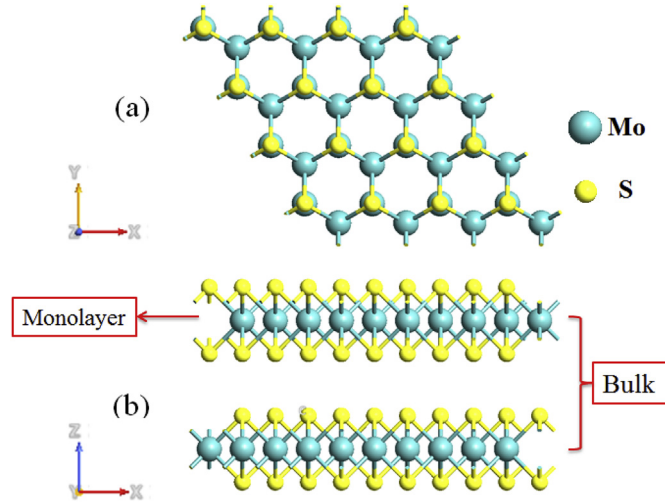
Inside the atomic spheres for wave function expansion, the maximum value of orbital quantum number  $l$  is confined to  $l_{max} = 10$ . In the interstitial region, the wave functions were extended regarding plane waves with the cut-off value of  $RK_{max} = 7$ . Besides, magnitude of largest vector in Fourier expansion of the charge density was set to  $G_{max} = 12$  (a.u.)<sup>-1</sup>. All these values have been chosen in a way to ensure the convergence of the results. These above-mentioned parameters are used further for the calculations of the electronic structure and optical properties of both bulk and monolayer MoS<sub>2</sub>.

For calculations of the exchange–correlation potential, generalized gradient approximation (GGA) by Perdew–Burke–Ernzerhof (PBE) [37,38] potential was used. Further, the tetrahedron method by Blochl et al. [39] was employed for integration through the Brillouin zone (BZ). The BZ sampling was made using 1000  $k$ -points within the full zone. The iteration process was verified accounting changes in the integral charge difference  $q = \int |\rho_n(r) - \rho_{n-1}(r)| dr$ , where  $\rho_{n-1}(r)$  and  $\rho_n(r)$  are input previous iteration and output current iteration charge density, respectively, and the calculations were intercepted when reaching the value  $q \leq 10^{-4}$ .

## 3. Results and discussion

Atomic structure of MoS<sub>2</sub> is shown in Fig. 1. It can be seen that MoS<sub>2</sub> crystal has the hexagonal structure, which consists of S–Mo–S layers. In each layer, the Mo atoms are arranged in the hexagonal lattice and positioned in trigonal prismatic coordination with the two S layers. After relaxation, the lattice constants of bulk and monolayer MoS<sub>2</sub> are 3.176 Å and 3.180 Å, respectively. This result is in good agreement with the previous theoretical calculations [25,31] and the experimental measurements [7,40]. Moreover, our obtained Mo–S bond length in both MoS<sub>2</sub> bulk and monolayer is 1.41 Å, which is consistent with the experimental value [7]. In Fig. 2 we display the electronic band structures of both MoS<sub>2</sub> bulk and monolayer. We found that MoS<sub>2</sub> bulk is a semiconductor with an indirect band gap, opening between the highest occupied energy state of the valence band located at the  $\Gamma$  point and the lowest unoccupied energy state of the conduction band located at the  $K$  point. When MoS<sub>2</sub> bulk is transformed to a monolayer, it becomes a semiconductor with a direct band gap, opening between the highest occupied state of the valence band and the lowest unoccupied state of the conduction band located at the  $K$  point. Our calculated band gaps for the bulk and monolayer MoS<sub>2</sub> are respectively 1.23 eV and 1.70 eV which are in good agreement with the available data [7,41].

We next focus on the optical properties of MoS<sub>2</sub> bulk and monolayer in the energy range from 0 to 15 eV. Real and imaginary parts of dielectric function for bulk and monolayer MoS<sub>2</sub> are calculated by considering interband transition for both directions of electric field: (i) parallel  $E_{\parallel x}$  and (ii) perpendicular  $E_{\parallel z}$ . The  $\epsilon_2(\omega)$  constituents are worth to be known for depiction of the linear optical susceptibility of the crystal and they are denoted as follows [42]:



**Fig. 1.** The top view (a) and side view (b) of the atomic structure of the MoS<sub>2</sub>. The green and yellow balls stand for the Mo and S atoms, respectively. (For interpretation of the references to colour in this figure legend, the reader is referred to the Web version of this article.)

$$\varepsilon_2^{ij}(\omega) = \frac{4\pi^2 e^2}{Vm^2 \omega^2} \sum_{nn'\sigma} \langle kn\sigma | p_i | kn'\sigma \rangle \langle kn'\sigma | p_j | kn\sigma \rangle \times f_{kn} (1 - f_{kn'}) \delta(E_{kn'} - E_{kn} - \hbar\omega), \quad (1)$$

where  $e$  and  $m$  stand respectively for the charge and mass of electron,  $\omega$  is the angular frequency of electromagnetic irradiation,  $V$  is a unit-cell volume,  $p$  is the momentum operator,  $|kn\sigma\rangle$  is the wave function of a crystal with crystal wave vector  $\vec{k}$ ,  $\sigma$  is spin which corresponds to the energy eigenvalue  $E_{kn}$ , and  $f_{kn}$  is the Fermi distribution function bringing a certain count to transitions from occupied states to unoccupied states.

The real part  $\varepsilon_1(\omega)$  and imaginary  $\varepsilon_2(\omega)$  parts of dielectric function  $\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega)$  for bulk and monolayer MoS<sub>2</sub> are obtained by the Kramers-Kronig transformation from its corresponding imaginary part as following [43]:

$$\varepsilon_1(\omega) = 1 + \frac{2}{\pi} P \int_0^{\infty} \frac{\omega' \varepsilon_2(\omega')}{\omega'^2 - \omega^2} d\omega', \quad (2)$$

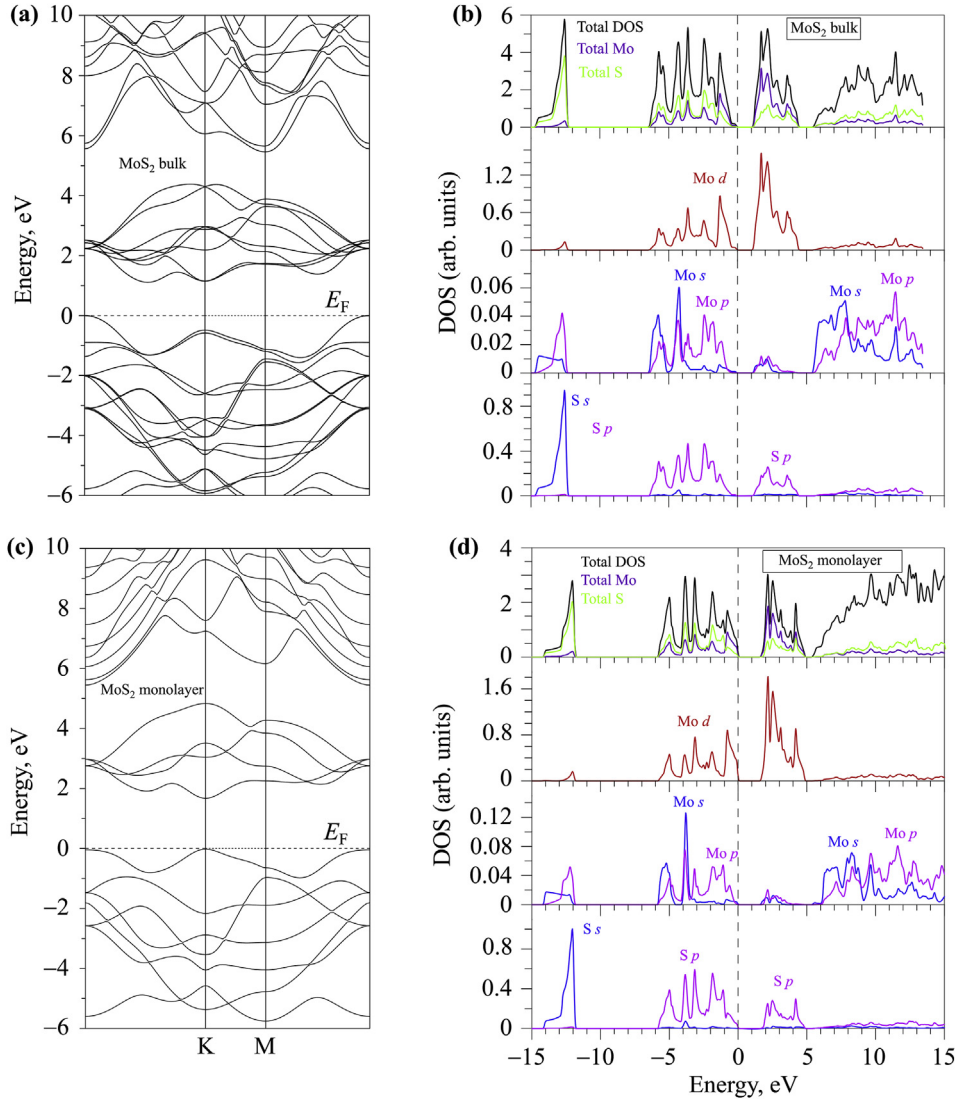
where  $P$  stands for the principal value of the integral.

In Fig. 3 we presented the real and imaginary parts of the dielectric function of both bulk and monolayer MoS<sub>2</sub>. We can see that the dielectric function for MoS<sub>2</sub> bulk is completely different from the dielectric function of MoS<sub>2</sub> monolayer and the values of dielectric constants are different in the out-plane and in-plane directions of the electric field. At equilibrium state the dielectric constant in the parallel  $E_{\parallel x}$  and perpendicular  $E_{\parallel z}$  directions is 15.01 and 8.92 for bulk MoS<sub>2</sub>, respectively. The dielectric constant is 4.95 and 2.92 for monolayer MoS<sub>2</sub> in  $E_{\parallel x}$  and  $E_{\parallel z}$ , respectively [see Fig. 3(a)]. Fig. 3 shows that, in the energy range from 0 to 6 eV, the dielectric functions have highly anisotropic, while they become isotropic when the energy is larger than 7 eV.

In Fig. 3(a,c), we also showed the dielectric functions of bulk MoS<sub>2</sub> in comparison with the experimental results. Our calculations for both real  $\varepsilon_1$  and imaginary  $\varepsilon_2$  parts of the dielectric constant of bulk MoS<sub>2</sub> are in good agreement with the previous experimental measurements [44]. The imaginary part of the dielectric function for bulk MoS<sub>2</sub> has three structure peaks A, B, and C (counts both  $E_{\parallel x}$  and  $E_{\parallel z}$ ) at the energy of 2.8 eV, 4.4 eV, and 5.5 eV, respectively, as shown in Fig. 3(c). While, the imaginary part of dielectric functions for MoS<sub>2</sub> monolayer shows four structure peaks A, B, C, and D at the energy of 2.9 eV, 3.7 eV, 4.4 eV, and 5.2 eV, respectively. It is shown that the additional structure peak B at the energy of 3.7 eV is appeared for monolayer MoS<sub>2</sub> in comparison with bulk MoS<sub>2</sub>, as shown in Fig. 3(d).

Fig. 4 shows the adsorption coefficient  $\alpha(\omega)$  and optical reflectivity  $R(\omega)$  of bulk and monolayer MoS<sub>2</sub> in both  $E_{\parallel x}$  and  $E_{\parallel z}$  directions. It is known that the optical absorption spectrum is vital in investigating the optical properties of MoS<sub>2</sub>. The absorption coefficient  $\alpha(\omega)$  and reflectance can be calculated as follows

$$\alpha^{ij}(\omega) = \frac{2\omega k^{ij}(\omega)}{c}, \quad (3)$$

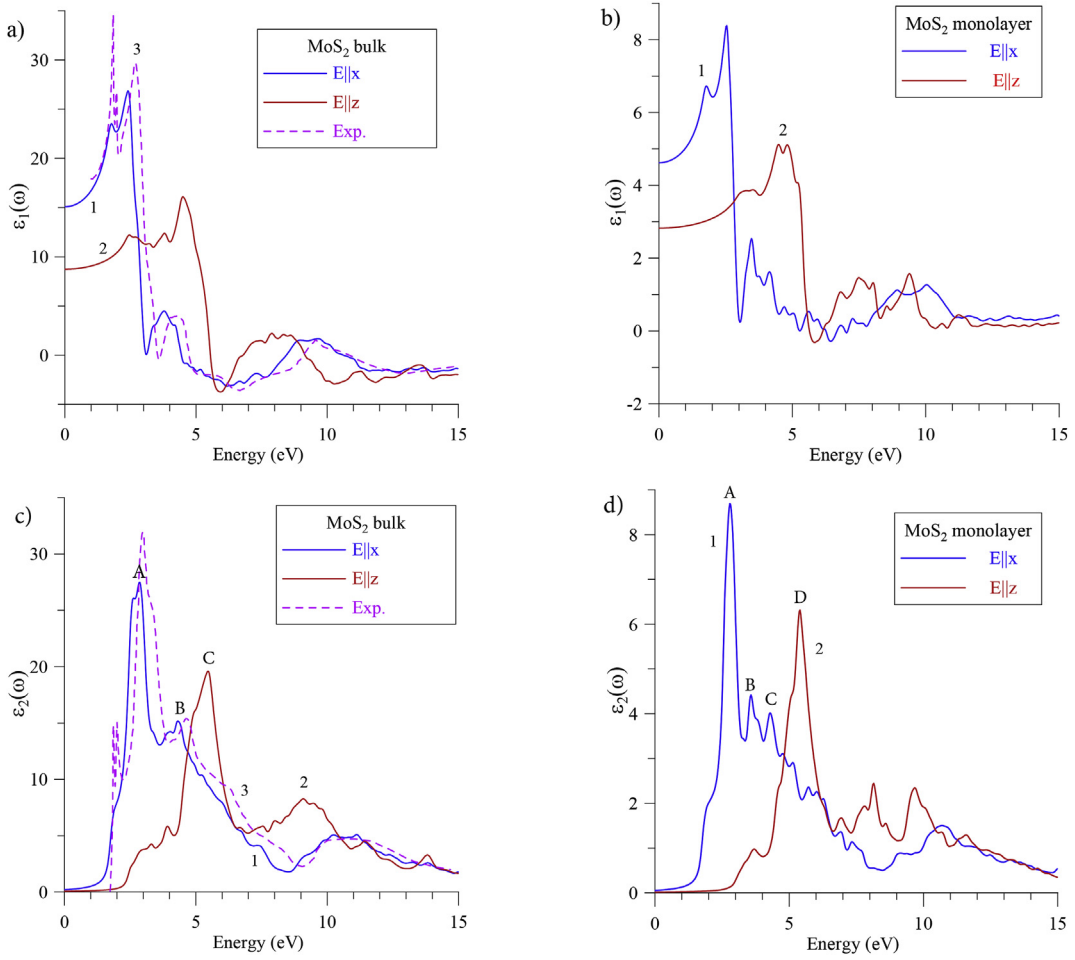


**Fig. 2.** Electronic band structure and density of states of the bulk MoS<sub>2</sub> (a) and (b) and monolayer MoS<sub>2</sub> (c) and (d), respectively. The Fermi energy level is set to 0 eV.

$$R^{ij}(\omega) = \frac{(n^{ij} - 1)^2 + k^{ij^2}}{(n^{ij} - 1)^2 - k^{ij^2}} = \left| \frac{\sqrt{\epsilon_1^{ij} + i\epsilon_2^{ij}} - 1}{\sqrt{\epsilon_1^{ij} + i\epsilon_2^{ij}} + 1} \right|^2. \quad (4)$$

As shown in Fig. 4(a), for bulk MoS<sub>2</sub> in  $E_{\parallel x}$  direction, the minimum of adsorption spectra is observed about 8.8 eV associated with the collective plasmons resonance, whereas the first and the highest peaks occur at 3 eV and 14 eV, respectively, which are related to  $\pi$ -electron plasmon peak. Also, for bulk MoS<sub>2</sub> in  $E_{\parallel z}$  direction, the minimum of adsorption spectra is observed about 7 eV associated with the collective plasmons resonance, whereas the first and the highest peaks occur at 3.4 eV and 14.2 eV which are related to  $(\pi + \sigma)$  electron plasmon peak. Besides, from this figure we can see that the transition in  $E_{\parallel z}$  direction is more than that in  $E_{\parallel x}$  direction. However, in comparison with the experiment result (the dashed line, which is adapted from the Ref. [44]) we see that the transition in  $E_{\parallel x}$  direction is accordant with the experiment data more than the transition in  $E_{\parallel z}$  direction is. This conclusion is also valid for optical reflectivity  $R(\omega)$  in Fig. 4(c), where we showed the dependence of  $R(\omega)$  on the energy.

Fig. 4(b) shows the adsorption spectra of monolayer MoS<sub>2</sub>, there is shown that the minimum of the adsorption spectra is observed around 8.6 eV. The experimental data of the adsorption coefficient and reflectance of bulk MoS<sub>2</sub> is also plotted for



**Fig. 3.** Real  $\varepsilon_1(\omega)$  and imaginary  $\varepsilon_2(\omega)$  parts of dielectric function of bulk (a,c) and monolayer (b,d) MoS<sub>2</sub> calculated by GGA approximation in the cases of  $E_{\parallel x}$  (1) and  $E_{\parallel z}$  (2). The experimental data from Ref. [44] is shown by the dashed lines (3) for the case of bulk MoS<sub>2</sub> with the parallel electric field direction for comparison.

comparison. In comparison with the experimental data [44], we can see that our calculated adsorption spectra for both bulk and monolayer MoS<sub>2</sub> are associated with the energy of about 9 eV.

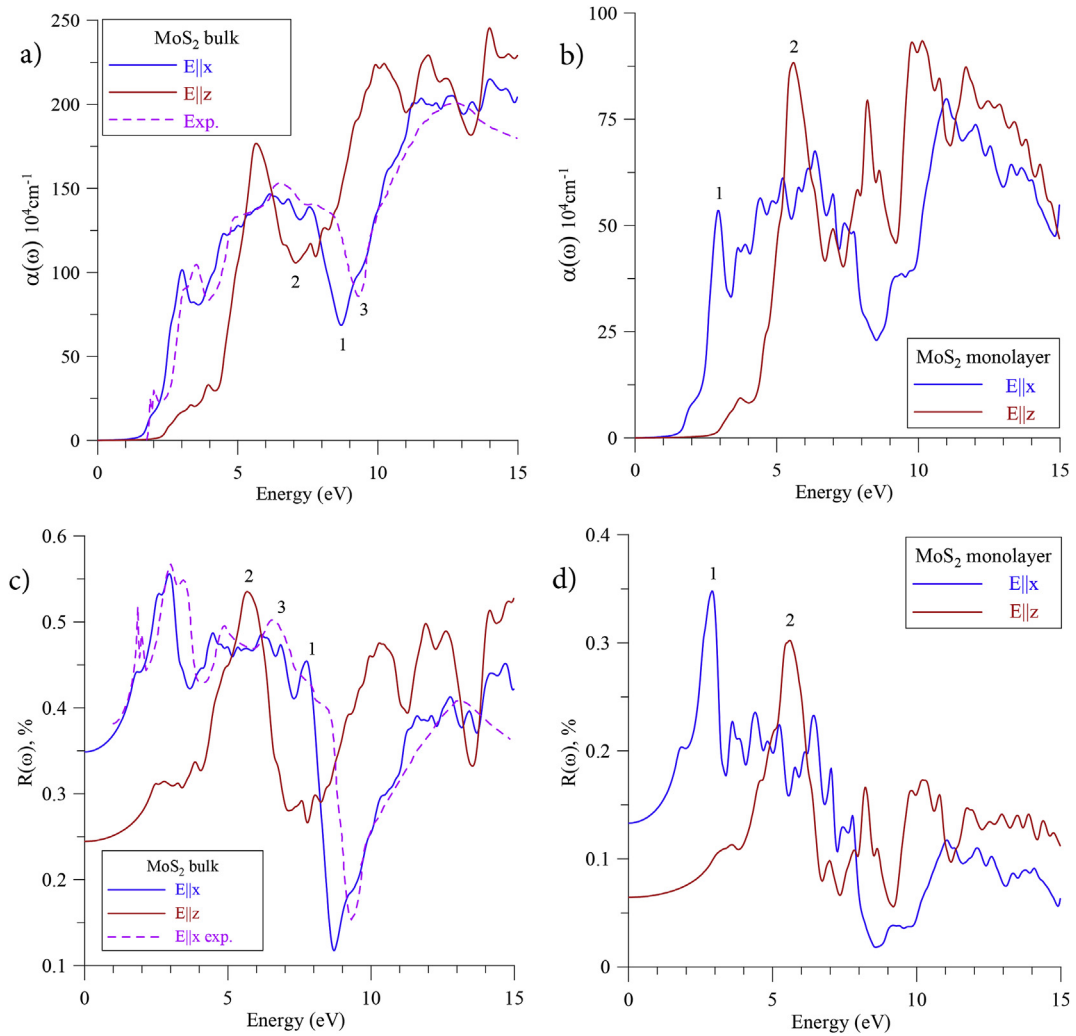
Fig. 5 shows the calculated refraction index  $n^{ij}(\omega)$  and extinction index  $k^{ij}(\omega)$  of bulk and monolayer MoS<sub>2</sub> under both directions of electric field. The refraction index and extinction index in both bulk and monolayer MoS<sub>2</sub> can be calculated by Ref. [45].

$$n^{ij}(\omega) = \frac{1}{\sqrt{2}} \left[ \sqrt{\varepsilon_1^{ij}(\omega)^2 + \varepsilon_2^{ij}(\omega)^2} + \varepsilon_1^{ij}(\omega) \right]^{1/2}, \quad (5)$$

$$k^{ij}(\omega) = \frac{1}{\sqrt{2}} \left[ \sqrt{\varepsilon_1^{ij}(\omega)^2 + \varepsilon_2^{ij}(\omega)^2} - \varepsilon_1^{ij}(\omega) \right]^{1/2}. \quad (6)$$

We can see that the static refraction index (the value of refraction coefficient at zero energy) in the  $E_{\parallel x}$  direction of electric field of bulk and monolayer MoS<sub>2</sub> is 3.88 and 2.15, respectively and this value in the  $E_{\parallel z}$  direction of electric field is 2.96 and 1.68, respectively.

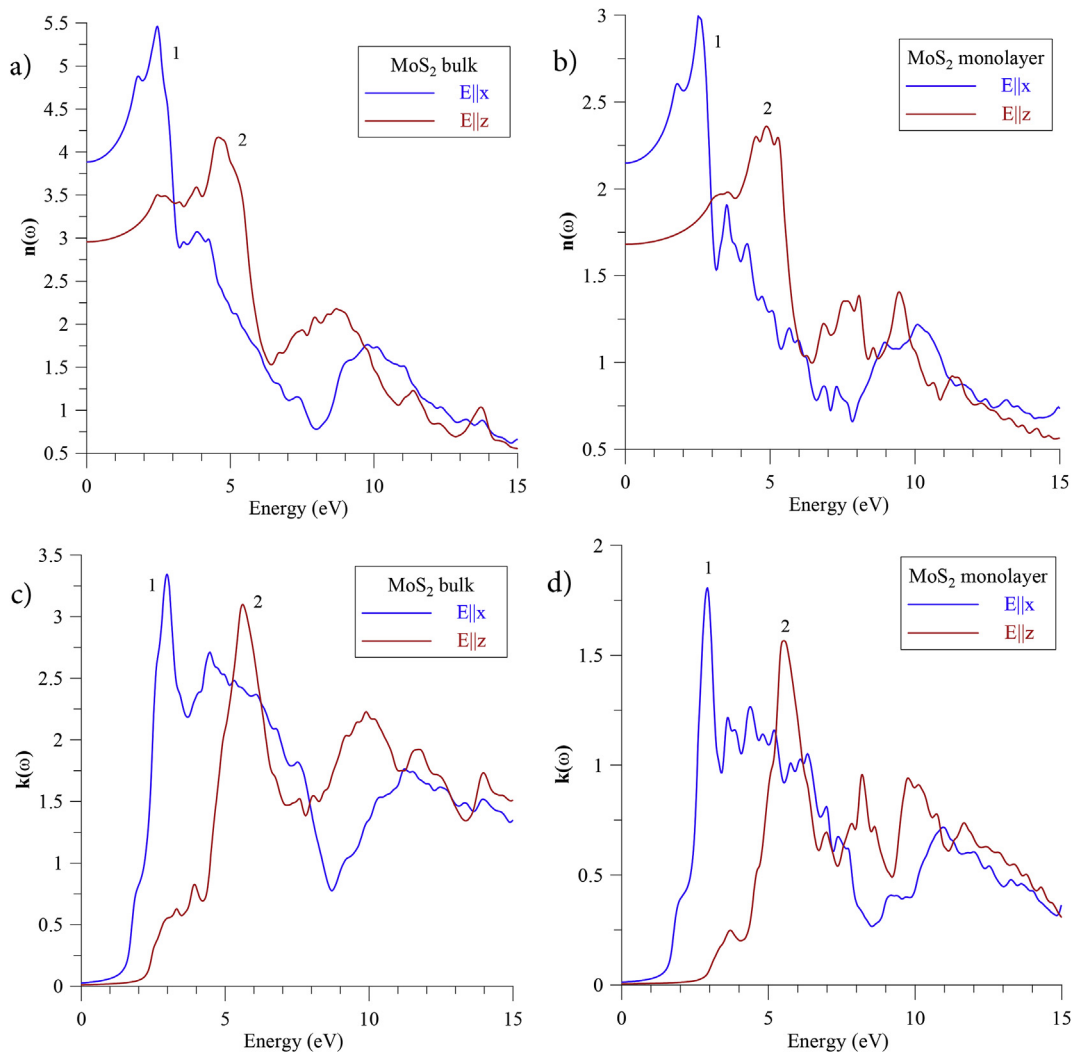
The electron energy loss spectrum for bulk and monolayer MoS<sub>2</sub> in both parallel and perpendicular electric field direction is illustrated in Fig. 6. We all know that the electron energy loss function is a key factor which describes the energy loss of a fast electron traversing in the material. The sharp energy loss peak attributes to the plasma resonance and the corresponding frequency is the so-called plasma frequency. The energy loss function in terms of real and imaginary parts of dielectric constant can be defined as follows:



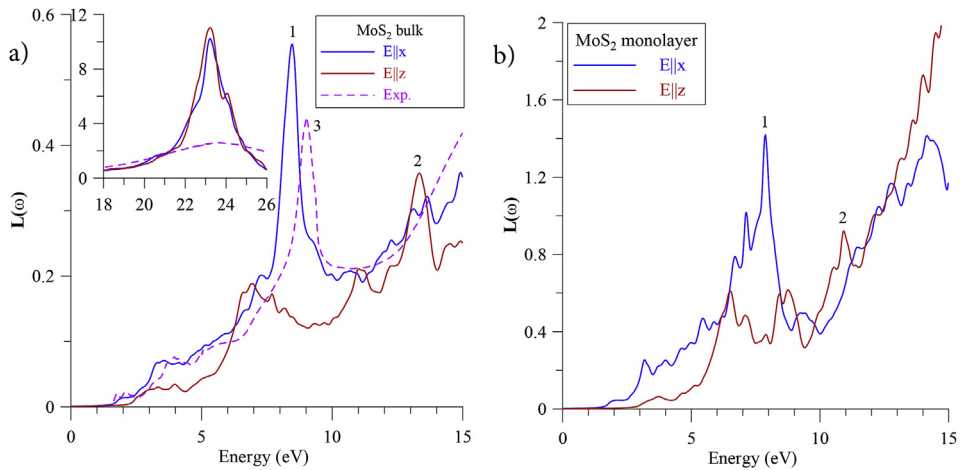
**Fig. 4.** Absorption coefficient  $\alpha(\omega)$  (a, b) and optical reflectivity  $R(\omega)$  (c, d) of bulk and monolayer MoS<sub>2</sub>, respectively. (1) and (2) stand for the cases of  $E_{||x}$  and  $E_{||z}$ , respectively. The dashed lines (3) in (a,c) are the experiment data from Ref. [44].

$$L^{ij}(\omega) = -\text{Im}(\epsilon^{-1})^{ij} = \frac{\epsilon_2^{ij}(\omega)}{\epsilon_1^{ij}(\omega)^2 + \epsilon_2^{ij}(\omega)^2}. \quad (7)$$

Fig. 6(a) shows the electron energy loss spectra for bulk MoS<sub>2</sub> in both perpendicular and parallel electric field directions. The experimental value is also plotted for comparison. As shown in Fig. 6(a), our calculated spectra shows the energy loss sharp peak of  $E_{||x}$  at 8.65 eV, which is consistent with the experimental value with energy loss peak at 8.81 eV [44]. In addition, we found a prominent peak at 23.1 eV in both parallel  $E_{||x}$  and perpendicular  $E_{||z}$  directions of electric field, as shown in the inset of Fig. 6(a). This value is consistent with experiment value of 23 eV in the energy loss spectra [44]. In Fig. 6(b) we show calculated electron energy loss spectra for both parallel  $E_{||x}$  and perpendicular  $E_{||z}$  direction of electric field in MoS<sub>2</sub> monolayer. It can be seen that for parallel  $E_{||x}$  electric field the sharp energy loss peak at 8.5 eV was observed in monolayer MoS<sub>2</sub>. It is in agreement with the energy loss peak in bulk MoS<sub>2</sub> due to the collective plasmonic oscillations. It is clear that the traditional DFT method underestimates the band gap of materials like as semiconductors and insulators. Thus, it should be noted that the different calculated methods like as PBE, or hybrid functional may give rise to different values in determining the electronic structure, relative variations of the electronic properties, leading to the change in the optical properties of these materials.



**Fig. 5.** Refractive index  $n(\omega)$  and extinction coefficient  $k(\omega)$  of MoS<sub>2</sub> bulk (a,c) and monolayer (b,d), respectively. (1) and (2) stand respectively for the cases of  $E_{||x}$  and  $E_{||z}$ .



**Fig. 6.** Electron energy-loss spectrum  $L(\omega)$  of bulk (a) and monolayer (b) MoS<sub>2</sub> in the cases of  $E_{||x}$  (1) and  $E_{||z}$  (2). The inset of (a) is the plot of  $L(\omega)$  in the larger range of energy. The dashed line (3) in (a) is the experiment data from Ref. [44].

#### 4. Conclusion

In conclusion, we study the optical properties of bulk and monolayer MoS<sub>2</sub> using DFT calculations. The obtained band gaps of 1.23 eV and 1.70 eV for the bulk and monolayer MoS<sub>2</sub>, respectively, are in good agreement with the available data. For the optical characters of MoS<sub>2</sub>, we investigated the dielectric function, optical reflectivity, and electron energy-loss spectrum in the energy range from 0 to 15 eV. Our DFT calculations showed that the difference in the dielectric constant between bulk and monolayer MoS<sub>2</sub> is significant in both directions of the electric field. The dielectric functions have highly anisotropic in the energy range from 0 to 6 eV but they become isotropic when the energy is larger than 7 eV. In comparison with bulk MoS<sub>2</sub>, the appearance of the peak at 3.7 eV in the imaginary part of the dielectric function of monolayer MoS<sub>2</sub> demonstrates that quantum confinement can increase the interband transition in the monolayer MoS<sub>2</sub>. The observed prominent peak at 23.1 eV in the energy-loss spectra agrees with experimental measurement. The obtained results may provide useful potential applications for the MoS<sub>2</sub> structures in electronic and optoelectronic devices.

#### Acknowledgments

This research is funded by Vietnam National Foundation for Science and Technology Development (NAFOSTED) under Grant Number 103.01-2016.07 and the Belarusian Scientific Program "Convergence".

#### References

- [1] K.S. Novoselov, A.K. Geim, S.V. Morozov, D. Jiang, Y. Zhang, S.V. Dubonos, I.V. Grigorieva, A.A. Firsov, Electric field effect in atomically thin carbon films, *Science* 306 (2004) 666.
- [2] F. Schwierz, Graphene transistors, *Nat. Nanotechnol.* 5 (2010) 487.
- [3] D.G. Papageorgiou, I.A. Kinloch, R.J. Young, Mechanical properties of graphene and graphene-based nanocomposites, *Prog. Mater. Sci.* 90 (2017) 75.
- [4] C. Neto, A.H.F. Guinea, N.M.R. Peres, K.S. Novoselov, A.K. Geim, The electronic properties of graphene, *Rev. Mod. Phys.* 81 (1) (2009) 109.
- [5] M. Sun, J.-P. Chou, Q. Ren, Y. Zhao, J. Yu, W. Tang, Tunable Schottky barrier in van der Waals heterostructures of graphene and g-GaN, *Appl. Phys. Lett.* 110 (17) (2017), 173105.
- [6] M. Sun, W. Tang, Q. Ren, S. Wang, Jin Yu, Y. Du, Y. Zhang, First-principles study of the alkali earth metal atoms adsorption on graphene, *Appl. Surf. Sci.* 356 (2015) 668.
- [7] K.F. Mak, C. Lee, J. Hone, J. Shan, T.F. Heinz, Atomically thin MoS<sub>2</sub>: a new direct-gap semiconductor, *Phys. Rev. Lett.* 105 (13) (2010), 136805.
- [8] Q. Tang, Z. Zhou, Z. Chen, Innovation and discovery of graphene-like materials via density-functional theory computations, in: *Wiley Interdisciplinary Reviews: Computational Molecular Science* vol. 5, 2015, p. 360.
- [9] Q. Tang, Z. Zhou, Graphene-analogous low-dimensional materials, *Prog. Mater. Sci.* 58 (2013) 1244.
- [10] M. Sun, W. Tang, Q. Ren, Y. Zhao, S. Wang, J. Yu, Y. Du, Y. Hao, Electronic and magnetic behaviors of graphene with 5d series transition metal atom substitutions: a first-principles study, in: *Physica E: Low-dimensional Systems and Nanostructures* vol. 80, 2016, p. 142.
- [11] L.-J. Kong, G.-H. Liu, L. Qiang, Electronic and optical properties of O-doped monolayer MoS<sub>2</sub>, *Comput. Mater. Sci.* 111 (2016) 416.
- [12] M. Sun, J.-P. Chou, J. Yu, W. Tang, Electronic properties of blue phosphorene/graphene and blue phosphorene/graphene-like gallium nitride heterostructures, *Phys. Chem. Chem. Phys.* 19 (26) (2017), 17324.
- [13] M. Sun, J.-P. Chou, Y. Zhao, J. Yu, W. Tang, Weak C-H...F-C hydrogen bonds make a big difference in graphene/fluorographane and fluorographene/fluorographane bilayers, *Phys. Chem. Chem. Phys.* 19 (41) (2017) 28127.
- [14] M. Sun, Q. Ren, Y. Zhao, J.-P. Chou, J. Yu, W. Tang, Electronic and magnetic properties of 4d series transition metal substituted graphene: a first-principles study, *Carbon* 120 (2017) 265.
- [15] D. Yang, S.J. Sandoval, W.M.R. Divigalpitiya, J.C. Irwin, R.F. Frindt, Structure of single-molecular-layer MoS<sub>2</sub>, *Phys. Rev. B* 43 (14) (1991), 12053.
- [16] M. Chhowalla, H.S. Shin, G. Eda, L.-J. Li, K.P. Loh, H. Zhang, The chemistry of two-dimensional layered transition metal dichalcogenide nanosheets, *Nat. Chem.* 5 (2013) 263.
- [17] X. Huang, Z. Zeng, H. Zhang, Metal dichalcogenide nanosheets: preparation, properties and applications, *Chem. Soc. Rev.* 42 (5) (2013) 1934.
- [18] Y. Cai, J. Lan, G. Zhang, Y.-W. Zhang, Lattice vibrational modes and phonon thermal conductivity of monolayer MoS<sub>2</sub>, *Phys. Rev. B* 89 (3) (2014), 035438.
- [19] Y. Cai, G. Zhang, Y.-W. Zhang, Polarity-reversed robust carrier mobility in monolayer MoS<sub>2</sub> nanoribbons, *J. Am. Chem. Soc.* 136 (17) (2014) 6269.
- [20] Y. Jing, X. Tan, Z. Zhou, P. Shen, Tuning electronic and optical properties of MoS<sub>2</sub> monolayer via molecular charge transfer, *J. Mater. Chem.* 2 (2014), 16892.
- [21] D. Kim, D. Sun, W. Lu, Z. Cheng, Y. Zhu, D. Le, T.S. Rahman, L. Bartels, Toward the growth of an aligned single-layer MoS<sub>2</sub> film, *Langmuir* 27 (18) (2011), 11650.
- [22] K.S. Novoselov, D. Jiang, F. Schedin, T.J. Booth, V.V. Khotkevich, S.V. Morozov, A.K. Geim, Two-dimensional atomic crystals, *Proc. Natl. Acad. Sci. U. S. A.* 102 (2005), 10451.
- [23] J.N. Coleman, M. Lotya, A. O'Neill, S.D. Bergin, P.J. King, U. Khan, K. Young, A. Gaucher, S. De, R.J. Smith, I.V. Shvets, S.K. Arora, G. Stanton, H.-Y. Kim, K. Lee, G.T. Kim, G.S. Duesberg, T. Hallam, J.J. Boland, J.J. Wang, J.F. Donegan, J.C. Grunlan, G. Moriarty, A. Shmeliov, R.J. Nicholls, J.M. Perkins, E.M. Grieveson, K. Theuwissen, D.W. McComb, P.D. Nellist, V. Nicolosi, Two-dimensional nanosheets produced by liquid exfoliation of layered materials, *Science* 331 (2011) 568.
- [24] Q.H. Wang, K. Kalantar-Zadeh, A. Kis, J.N. Coleman, M.S. Strano, Electronics and optoelectronics of two-dimensional transition metal dichalcogenides, *Nat. Nanotechnol.* 7 (2012) 699.
- [25] C. Ataca, M. Topsakal, E. Aktürk, S. Ciraci, A comparative study of lattice dynamics of three- and two-dimensional MoS<sub>2</sub>, *J. Phys. Chem. C* 115 (33) (2011), 16354.
- [26] C.V. Nguyen, N.N. Hieu, Effect of biaxial strain and external electric field on electronic properties of MoS<sub>2</sub> monolayer: a first-principle study, *Chem. Phys.* 468 (2016) 9.
- [27] C.V. Nguyen, N.N. Hieu, D.T. Nguyen, Dispersion-corrected density functional theory investigations of structural and electronic properties of bulk MoS<sub>2</sub>: effect of uniaxial strain, *Nanoscale Res. Lett.* 10 (1) (2015) 433.
- [28] P. Johari, V.B. Shenoy, Tuning the electronic properties of semiconducting transition metal dichalcogenides by applying mechanical strains, *ACS Nano* 6 (2012) 5449.
- [29] J.-Y. Noh, H. Kim, Y.-S. Kim, Stability and electronic structures of native defects in single-layer MoS<sub>2</sub>, *Phys. Rev. B* 89 (2014), 205417.
- [30] A. Kumar, P. Ahluwalia, Tunable dielectric response of transition metals dichalcogenides MX<sub>2</sub> (M=Mo, W; X=S, Se, Te): effect of quantum confinement, *Phys. B Condens. Matter* 407 (24) (2012) 4627.
- [31] A. Kumar, P. Ahluwalia, A first principle comparative study of electronic and optical properties of 1H-MoS<sub>2</sub> and 2H-MoS<sub>2</sub>, *Mater. Chem. Phys.* 135 (2012) 755.



- [32] W.B. Xu, B.J. Huang, P. Li, F. Li, C.W. Zhang, P.J. Wang, The electronic structure and optical properties of Mn and B, C, N co-doped MoS<sub>2</sub> monolayers, *Nanoscale Res. Lett.* 9 (2014) 554.
- [33] A. Molina-Sánchez, K. Hummer, L. Wirtz, Vibrational and optical properties of MoS<sub>2</sub>: from monolayer to bulk, *Surf. Sci. Rep.* 70 (4) (2015) 554.
- [34] L.-p. Feng, J. Su, Z.-t. Liu, Effect of vacancies on structural, electronic and optical properties of monolayer MoS<sub>2</sub>: a first-principles study, *J. Alloy. Comp.* 613 (2014) 122.
- [35] C.V. Nguyen, N.N. Hieu, N.A. Poklonski, V.V. Ilyasov, L. Dinh, T.C. Phong, L.V. Tung, H.V. Phuc, Magneto-optical transport properties of monolayer MoS<sub>2</sub> on polar substrates, *Phys. Rev. B* 96 (12) (2017) 125411.
- [36] P. Blaha, K. Schwarz, G.K.H. Madsen, D. Kvasnicka, J. Luitz, WIEN2k: An Augmented Plane Wave Plus Local Orbitals Program for Calculating Crystal Properties, Vienna University of Technology, Austria, 2001.
- [37] J.P. Perdew, K. Burke, M. Ernzerhof, Generalized gradient approximation made simple, *Phys. Rev. Lett.* 77 (18) (1996) 3865.
- [38] J.P. Perdew, K. Burke, M. Ernzerhof, Generalized gradient approximation made simple [*Phys. Rev. Lett.* 77, 3865 (1996)], *Phys. Rev. Lett.* 78 (7) (1997) 1396.
- [39] P.E. Blöchl, O. Jepsen, O.K. Andersen, Improved tetrahedron method for brillouin-zone integrations, *Phys. Rev. B* 49 (23) (1994), 16223.
- [40] J. Wilson, A. Yoffe, The transition metal dichalcogenides discussion and interpretation of the observed optical, electrical and structural properties., *Adv. Phys.* 18 (1969) 193.
- [41] H.S.S. Ramakrishna Matte, A. Gomathi, A.K. Manna, D.J. Late, R. Datta, S.K. Pati, C.N.R. Rao, MoS<sub>2</sub> and WS<sub>2</sub> analogues of graphene, *Angew. Chem. Int. Ed.* 49 (24) (2010) 4059.
- [42] A. Delin, P. Ravindran, O. Eriksson, J. Wills, Full-potential optical calculations of lead chalcogenides, *Int. J. Quant. Chem.* 69 (3) (1998) 349.
- [43] H. Tributsch, Solar energy-assisted electrochemical splitting of water. some energetical, kinetical and catalytical considerations verified on MoS<sub>2</sub> layer crystal surfaces, *Zeitschrift für Naturforschung A* 32 (9) (1977) 972.
- [44] A. Beal, H. Hughes, Kramers-kronig analysis of the reflectivity spectra of 2H-MoS<sub>2</sub>, 2H-MoSe<sub>2</sub> and 2H-MoTe<sub>2</sub>, *J. Phys. C Solid State Phys.* 12 (5) (1979) 881.
- [45] P. Ravindran, A. Delin, B. Johansson, O. Eriksson, J.M. Wills, Electronic structure, chemical bonding, and optical properties of ferroelectric and anti-ferroelectric NaNO<sub>2</sub>, *Phys. Rev. B* 59 (3) (1999) 1776.