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## Surface Science





# First principle study on the electronic properties and Schottky contact of graphene adsorbed on MoS<sub>2</sub> monolayer under applied out-plane strain



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#### ABSTRACT

In the present work, electronic properties and Schottky contact of graphene adsorbed on the MoS<sub>2</sub> monolayer under applied out-plane strain are studied using density functional theory calculations. Our calculations show that weak van derpp Waals interactions between graphene and monolayer  $MoS_2$  are dominated at the interlayer distance of 3.34 Å and the binding energy per C atom of -25.1 meV. A narrow band gap of 3.6 meV has opened in G/MoS<sub>2</sub> heterointerface, and it can be modulated by the out-plane strain. Furthermore, the Schottky barrier and Schottky contact types in the  $G/MoS_2$  heterointerface can be controlled by the out-plane strain. At the equilibrium state (d = 3.34 Å), the intrinsic electronic structure of G/MoS<sub>2</sub> heterointerface is well preserved and forms an *n*type Schottky barrier of 0.49 eV. When the interlayer distance decreases, the transition from n-type to p-type Schottky contact occurs at d = 2.74 Å. Our studies promote the application of ultrathin G/MoS<sub>2</sub> heterointerface in the next-generation nanoelectronic and photonic devices such as van-der-Waals-based field effect transistors.

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## 1. Introduction

Graphene is two-dimensional (2D) material with one-atom thickness and high carrier mobility [1,2]. However, graphene is a gapless semiconductor that causes the biggest problem for applications to graphenebased electronic devices [3,4]. Furthermore, electronic properties of graphene are very sensitive to external conditions such as impurities [5,6], external electric field [7] and interaction between graphene and substrates [7,8]. There exist several ways to open the band gap of graphene. One of them is mixing or hybridization the electronic states at *K* and *K'* by breaking the translational symmetry at Dirac point of  $\pi$ and  $\pi^*$  energy bands. An other way is breaking the sublattice symmetry in graphene. A simple way to control and modulate the band gap of graphene is deposition of the graphene on a substrate. Moreover, it is clear that the substrates are an essential component in graphene-based electronic devices.

The weak van-der-Waals (vdW) interaction between graphene and 2D semiconductor substrates has also been widely studied both theoretically and experimentally in several systems, such as G/MoS<sub>2</sub> [9], G/SiC [10], G/h-BN [8,11], G/α-Al<sub>2</sub>O<sub>3</sub> [12], and G/P [13]. Varchon and co-workers studied the possibility of opening a band gap at the Dirac point in the system of graphene deposited on SiC substrate [14]. Besides, the effect of the substrate on electronic properties of graphene has been investigated experimentally [15]. G. Giovannetti et. al shown that the graphene/h-BN heterointerface has a band gap of 53 meV [8]. It means that the band gap at the Dirac points of graphene is effectively modulated by the interaction with the *h*-BN substrate. Moreover, Huang's group has demonstrated that the graphene became a semiconductor with a band gap of about 180 meV when it locates on the Al<sub>2</sub>O<sub>3</sub> substrate [16]. Recently, the interaction between graphene and MoS<sub>2</sub> monolayer has also been studied theoretically [9,17,31]. In addition, the G/MoS<sub>2</sub> interface was successfully synthesized experimentally [18-20].

It is noted that MoS<sub>2</sub> monolayer is a semiconductor with a direct band gap of 1.80 eV [21,22]. The band gap is opened between the lowest energy of the conduction band and the highest energy of the valence band at the K-point. Moreover, MoS2 monolayer has been successfully fabricated [23] and widely studied both experimentally [21,24] and theoretically [25-29] due to its extraordinary properties, for example, high on/off current ratio of about 10<sup>8</sup> [30], and high room-temperature carrier mobility of 200 cm<sup>2</sup>/Vs [30]. These characters make  $MoS_2$  monolayers a promising material for applications in nanoelectronic and optoelectronic devices.

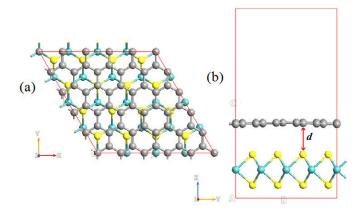
Interestingly, metal/semiconductor interface, such as G/MoS<sub>2</sub> interface, is a new contact type of the graphene-based vdW interfaces. This vdW interface shows much more novel properties than their unique

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**Fig. 1.** Atomic structure of  $G/MoS_2$ . Top (a), side (b) views of relaxed atomic structure of  $G/MoS_2$  interface. *d* is the interlayer distance between graphene and  $MoS_2$ . The blue, violet, and yellow balls stand for Mo, S, and C atoms, respectively. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

structures and the Schottky contact can be formed at the  $G/MoS_2$  interface. Also, it can be controlled by an external electric field and in-plane strain [31–33]. Liu and co-workers showed that the Schottky barrier of  $G/MoS_2$  heterointerface depends linearly on the electric field intensity [31]. In another work, they showed that the Schottky barrier can be controlled by the vertical electric field at the G/antimonene interface [32].

In this work, we investigate the electronic properties of the interaction between graphene and  $MoS_2$  monolayer (G/MoS\_2 interface) using first-principles calculations with vdW correction. The interaction between graphene and  $MoS_2$  monolayer is described by a weak vdW interaction at the equilibrium state. The dependence of band structures of G/MoS\_2 interface on the interlayer distances and the formation of Schottky contact at the G/MoS\_2 interface have also investigated. Moreover, the coupling between layers can be used to effectively control the Schottky barrier (*n*-type and *p*-type). Our results may prove some applications in vdW-based field effect transistors (FETs).

#### 2. Computational model and method

Our DFT calculations are performed using the plane wave basic as implemented in Quantum Espresso simulation package [34]. The ionelectron interaction is described by using the projector-augmented plane wave (PAW) approximation. The generalized gradient approximation (GGA) with the Perdew–Burke–Ernzerhof (PBE) functional is used for geometry optimization. The cut-off energy of 500 eV is adopted in our calculations. The Brillouin zone was represented by Monkhorst-Pack mesh of  $(9 \times 9 \times 1)$  is used for electronic band structure. It is known that the weak interactions are out of the framework of standard PBE functional. Thus we add the DFT-D2 (where D stands for dispersion) method with the Grimme vdW correction [35] to describe the weak interactions. Previously, we have used successfully this method for calculations of layered materials, such as graphene nanoribbon,  $MoS_2$  or graphene placed on semiconductor substrates [36–40].

The scheme of G/MoS<sub>2</sub> interface is shown in Fig. 1. The lattice parameters of the pristine graphene and the MoS<sub>2</sub> monolayer are 2.46 Å and 3.18 Å, respectively, which are in good agreement with the previous theoretical and experimental results [22,36,41,42]. Thus, we choose MoS<sub>2</sub> monolayer as a substrate to match well with graphene. We also use a supercell, which consists (5 × 5) unit cells of graphene and (4 × 4) unit cells of MoS<sub>2</sub> monolayer along the *x* and *y* directions. The lattice mismatch between graphene and MoS<sub>2</sub> monolayer, therefore, is only about 2 %. A vacuum space of 22 Å in the *z* direction is used to avoid the interaction between neighbor slabs.

The binding energy per carbon atom between graphene and MoS<sub>2</sub> monolayer is calculated by  $E_b = -[E_{G/MOS_2} - (E_G + E_{MOS_2})]/N_C$ , where  $E_{G/MOS_2}$ ,  $E_G$ , and  $E_{MOS_2}$  are the total energy of the G/MoS<sub>2</sub> interface, freestanding graphene, and isolated MoS<sub>2</sub> monolayer, respectively.  $N_C$  is the number of carbon atoms in the supercell.

### 3. Results and discussion

Firstly, we calculate the electronic band structures of freestanding graphene and  $MoS_2$  monolayer in the supercells using DFT-D2 calculations, as shown in Fig. 2(a and b). Our calculations show that the freestanding graphene has zero-gap [Fig. 2(a)] and monolayer  $MoS_2$  is a semiconductor with a direct band gap of 1.70 eV [Fig. 2(b)]. These results are in good agreement with those obtained in previous theoretical work [43]. We now investigate the electronic properties of the interface between graphene and monolayer  $MoS_2$  (G/MoS<sub>2</sub>) after full relaxation. It is noted that the lattice mismatch between graphene and  $MoS_2$  monolayer is small, about 2%. This small lattice mismatch has little affects on their electronic properties, and thus it does not affect the main results.

To obtain the equilibrium state, the G/MoS<sub>2</sub> interface is relaxed by using the DFT-D2 method. The relaxed atomic structure of G/MoS<sub>2</sub> interface is shown in Fig. 1. At equilibrium state, the interlayer distance between graphene and MoS<sub>2</sub> layers is  $d_0 = 3.34$  Å and the binding energy per carbon atom in G/MoS<sub>2</sub> interface is  $E_b = -25.1$  meV. These values are close to the previous theoretical and experimental results in graphene/substrate interface [8,44–46]. The negatible binding energy implies that the interaction between graphene and MoS<sub>2</sub> monolayer is a weak vdW and electrostatic interactions.

The electronic band structure of the  ${\rm G}/{\rm MoS}_2$  interface is shown in Fig. 2(c). We can see that the linear Dirac-like dispersion relation near the Fermi level of graphene is still preserved in G/MoS<sub>2</sub> interface. Magnifying the linear band structure, we can see that a band gap in G/MoS<sub>2</sub> interface is opened at the Dirac K-point of graphene [see the inset in Fig. 2(c)]. It means that at the Dirac *K*-point, the  $\pi$  and  $\pi^*$  bands repulse each other, forming a small energy band gap. Therefore, the G/MoS<sub>2</sub> interface is a semiconductor with a narrow direct energy gap, describing the effect of MoS<sub>2</sub> substrate on the electronic properties of graphene. Thus, the MoS<sub>2</sub> substrate can be used as an ideal substrate for graphenebased devices. The opened band gap of G/MoS<sub>2</sub> interface, as shown in the inset in Fig. 2(c), is 3.6 meV. This gap is smaller than that of G/h-BN [8,47,48], G/ZnO [46,49], and G/P [13,50,51]. To understand the physical nature of the band gap opening in G/MoS<sub>2</sub> interface, we use the tight-binding (TB) model analysis. According to the  $\pi$ -electron TB approximation of graphene, the dispersion relation near the Fermi level can be approximated as:

$$E(k) = \pm \sqrt{\Delta^2 + (\hbar v_F k)^2}$$

where *k* is the wave vector relative to Dirac point,  $v_F$  is the Fermi velocity, and  $\Delta$  is the on-site energy difference between the two sub-lattices of graphene. The signs correspond to conduction bands and valence bands, respectively. For freestanding graphene monolayer, the onsite energies of the two sub-lattices are identical ( $\Delta = 0$ ), resulting in the zero band gap and the linear dispersion relation near the Dirac point. For the G/MoS<sub>2</sub> heterointerface, the charge redistribution breaks the equivalence of the two graphene sub-lattices, giving rise to a non zero band gap,  $E_g = 2\Delta$ .

We now investigate the effect of out of plane strain on the electronic property of  $G/MOS_2$  interface. The out of plane strain is applied by changing the interlayer distance as  $\varepsilon = (d - d_0)/d_0$ , where d and  $d_0$  are the interlayer distances under applied strain and at equilibrium state, respectively. The interlayer distance varies from 2.44 Å to 4.24 Å along the z-direction. It means that the out of plane strain varies from -27% to 27%. In Fig. 3 we show the band structures of  $G/MOS_2$  interface at different interlayer distances. We can see that with increasing the interlayer distance, the conduction bands at Dirac point move downward to the Fermi energy level, whereas the valence bands shift downward

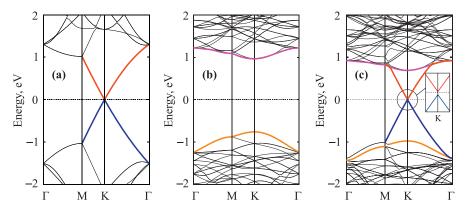
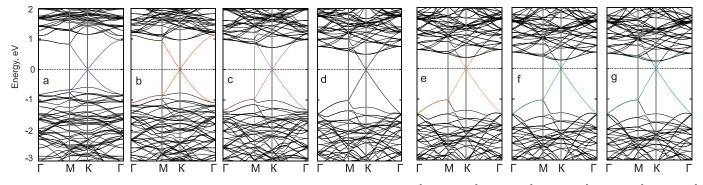
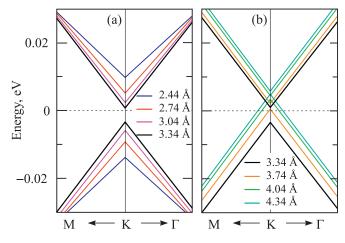


Fig. 2. DFT-D2 calculations of the band structure of freestanding graphene (a), substrate MoS<sub>2</sub> (b), and coupling  $G/MoS_2$  interface (c). The blue, and red lines stand respectively for bonding  $\pi$ -band and antibonding  $\pi$ <sup>\*</sup>-band of graphene. The purple, and orange lines stand for the conduction band minimum and the valence band maximum of isolated MoS<sub>2</sub> monolayer, respectively. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)



**Fig. 3.** Electronic energy band structure of G/MoS<sub>2</sub> interface with different interlayer distances *d*: (a) d = 2.44 Å, (b) d = 2.74 Å, (c) d = 3.04 Å, (d) d = 3.34 Å, (e) d = 3.64 Å, (f) d = 3.94 Å, and (g) d = 4.24 Å.



**Fig. 4.** Linear dispersion near the Fermi level at Dirac point of  $G/MoS_2$  interface at the different interlayer distances *d*: (a)  $d \le d_0$ , (b)  $d \ge d_0$ .

from Fermi energy level. However, in all case of interlayer distances, the Dirac cone always locates at the *K*-point. It can be concluded that the electronic structure around the Dirac cone of graphene is almost not affected by the interlayer distance.

To describe the effect of the interlayer distance on the band gap of  $G/MoS_2$  interface, in Fig. 4 we show the energy dispersion near the Fermi energy level at Dirac point for the valence band maximum (VBM) and the conduction band minimum (CBM) in the band structures of  $G/MoS_2$  interface at different interlayer distances. In the case of  $d > d_0$ , when the interlayer distance increases, the bonding  $\pi$ -bands tend to go up to the Fermi level [see Fig. 4(b)]. The effect of the interlayer distance

on the band gap and total energy is shown in Fig. 5. We can see that, at a large interface distance  $d \ge 3.74$  Å, the Dirac point of graphene moves up to the Fermi level, leading to the *p*-type (hole) doping in graphene in the graphene/MoS<sub>2</sub> interface, as shown in Fig. 4(b). As shown in Fig. 5(a), the total energy of G/MoS<sub>2</sub> interface is minimum at the equilibrium state of  $d_0 = 3.34$  Å. Therefore, it can be concluded that the band gap of graphene/MoS<sub>2</sub> interface depends strongly on the interlayer distance, especially when  $d < d_0$ .

Interestingly, in this interface, we found a Schottky contact, which has been formed between the metallic graphene and the semiconducting MoS<sub>2</sub> substrate. Our calculations show that the Schottky barrier can be controlled by the interlayer distance between the layers. We show the dependence of the Schottky barrier on the interlayer distance in Fig. 6. Based on the Schottky-Mott model [52] at the metal/semiconductor interface [53], the *n*-type Schottky barrier is defined by  $\Phi_{B,n} = E_C - E_F$ , where  $E_C$  is the conduction band minimum (CBM) and  $E_F$  is the Fermi energy level, respectively. The *p*-type Schottky barrier is defined by  $\Phi_{B,p}$  =  $E_V - E_F$ , where  $E_V$  is the valence band maximum (VBM). Note that the sum of the *n*-type and *p*-type Schottky barrier is approximately equal to the band gap of the semiconductor, that is  $\Phi_{Bn} + \Phi_{Bp} \approx E_g(MoS_2)$ . At the equilibrium state, the Fermi level locates at the middle of the band gap between the CBM and VBM bands. In Fig. 6, we plot the evolution of the *n*-type  $\Phi_{Bn}$  and *p*-type  $\Phi_{Bp}$  Schottky contact. It can be seen that the system at the equilibrium state shows an *n*-type Schottky contact with the barrier of  $\Phi_{B,n} = 0.49$  eV. With increasing the interlayer distance to d = 4.24 Å ( $\epsilon = 27\%$ ), the system still preserves *n*-type Schottky contact. The Schottky barrier decreases from 0.49 eV down to 0.28 eV and to 0.24 eV with the increasing distance from 3.34 Å ( $\epsilon = 0\%$ ) to 3.94 Å ( $\epsilon = 18\%$ ) and to 4.24 Å ( $\epsilon = 27\%$ ), respectively. This behavior can be explained as follows: when the interlayer distance increases, chemical interactions in the G/MoS<sub>2</sub> interface becomes weaker, resulting in

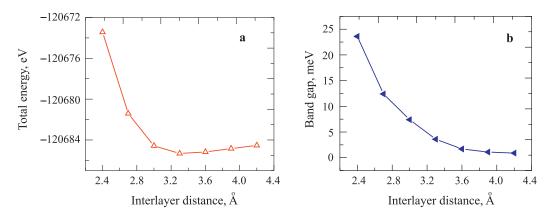


Fig. 5. Dependence of the total energy (a) and band gap (b) of G/MoS<sub>2</sub> interface on the interlayer distance.

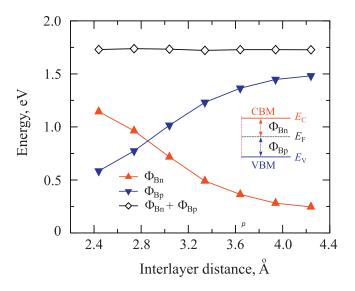
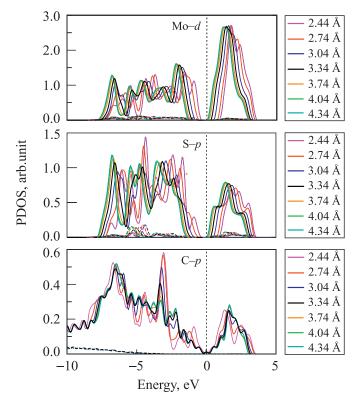


Fig. 6. Dependence of the Schottky barrier in G/MoS<sub>2</sub> interface on the interlayer distance.

fewer electrons transfer from MoS<sub>2</sub> to graphene. With the decrease of the interlayer distance to 3.04 Å ( $\varepsilon = -9\%$ ), the system shows a *n*-type Schottky contact with the barrier height of  $\Phi_{B,n} = 0.72$  eV. However, when the interlayer distance decreases to 2.74 Å ( $\varepsilon = -18\%$ ) and 2.44 Å ( $\varepsilon = -27\%$ ), the Schottky barrier is transformed from *n*-type to *p*-type Schottky contact. The transition of the Schottky barrier is located at the interlayer distance of 2.74 Å with the Schottky barrier of  $\Phi_{B,p} = 0.77$  eV. The Schottky barrier  $\Phi_{B,p}$  decreases to 0.51 eV at the interlayer distance of 2.44 Å. Thus, the interlayer distance not only controls the Schottky barrier but also causes the Schottky barrier transition from *n*-type to *p*-type Schottky contact.

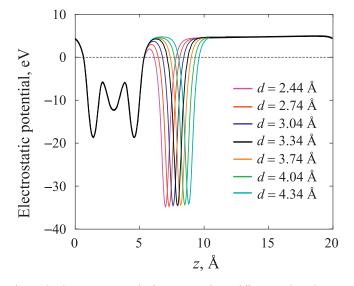
It is clear that the G/MoS<sub>2</sub> interface becomes an unified electronic system as the contact between graphene and MoS<sub>2</sub> monolayer, resulting in the appearance of a small band bending. The band bending can be defined as  $\Delta W = W_{G/MoS_2} - W_{MoS_2}$ , where  $W_{G/MoS_2}$  is the work function of the G/MoS<sub>2</sub> interface and  $W_{MoS_2}$  is the work function of the isolated MoS<sub>2</sub> monolayer. The calculated work function of the G/MoS<sub>2</sub> interface and isolated MoS<sub>2</sub> monolayer at the equilibrium state is 4.80 eV and 4.92 eV, respectively. Thus, the band bending at the equilibrium state is only -0.12 eV, which implies a n-type Schottky contact in the G/MoS<sub>2</sub> interface. By increasing the interlayer distance *d* from 3.34 Å to 4.24 Å, the work function of the G/MoS<sub>2</sub> interface decreases from 4.80 eV to 4.70 eV, respectively, resulting in a decrease of the band bending from -0.12 eV to -0.22 eV, respectively. In contrast, by decreasing the interlayer distance the band bending increases. For example, the band



**Fig. 7.** Partial density of states (PDOS) of Mo-*d* (a), S-*p* (b) and C-*p* (c) orbitals in the  $G/MoS_2$  systems with different interlayer distances.

bending is 4.91 eV and 4.95 eV at the interlayer distance of 3.04 Å and 2.74 Å, respectively. It can be seen that at the interlayer distance of 2.74 Å, the band bending is greater than zero ( $\Delta W = 0.03$  eV), which presents a p-type Schottky contact.

To have a more explicit view of the effect of the interlayer distance on the electronic properties of  $G/MoS_2$  interface, we plot the partial density of states (PDOS) for the  $C-p_z$ , Mo-*d*, and S-*p* orbitals in the interface, as shown in Fig. 6. It shows that the  $G/MoS_2$  interface is a semiconductor at the equilibrium state with the Fermi level located between the VBM and the CBM. Both the VBM and the CBM are formed by the C-*p*, S-*p* and Mo-*d* states. By changing the interlayer distance between graphene and  $MoS_2$ monolayer, the S-*p* and Mo-*d* states are upshifted or/and downshifted, as shown in Fig. 7. With increasing the interlayer distance from 3.34 Å to 4.24 Å, the semiconductor-metal transition has been observed due to existence of the VBM at the Fermi level.



**Fig. 8.** The electrostatic potentials of  $G/MoS_2$  interface at different interlayer distances *d* along *z*-direction. Note that the  $MoS_2$  layer in interface is fixed and only the graphene layer is moved away from the  $MoS_2$  layer.

Fig. 8 shows the electrostatic potential of G/MoS<sub>2</sub> interface along the z-direction at different interlayer distances. We found that the energy level of graphene is downshifted to the valence band of MoS<sub>2</sub> with decreasing the interlayer distance from 4.24 Å to 2.44 Å. Our work function of freestanding graphene of 4.4 eV is in good agreement with the experimental data of 4.514 eV [54]. Also, the work function of MoS<sub>2</sub> monolayer is 4.9 eV. The transition from *n*-type to *p*-type Schottky contact at d = 2.74 Å can be explained by the transfer of electrons from MoS<sub>2</sub> to graphene layer. It indicates that the electrons are likely to move into graphene layer by decreasing the interlayer distance. It also can be seen from Fig. 8, the vdW interaction at d = 4.24 Å is weak and the effect of graphene on MoS<sub>2</sub> and vice versa is negligible. In this case, the electrostatic potential well of MoS<sub>2</sub> is much higher than that of graphene. When the interlayer distance decreases, the tunneling energy barrier at interface gradually shifts downward so that more electrons get favored to transfer from MoS<sub>2</sub> layer to graphene.

#### 4. Conclusion

In conclusion, the electronic properties of G/MoS<sub>2</sub> interface have been studied using density functional theory with added vdW correction method. Our calculations show that at the equilibrium state, the interlayer distance between graphene and MoS<sub>2</sub> substrate is  $d_0 = 3.34$  Å and thus it is characterized by the vdW interaction with the binding energy per carbon atom of -25.1 meV. In the presence of MoS<sub>2</sub> substrate, graphene has a small band gap of 3.6 meV. This band gap depends strongly on the interlayer distance *d* when  $d < d_0$ . At the equilibrium state, the graphene/MoS<sub>2</sub> interface has *n*-type Schottky contact with the Schottky barrier  $\Phi_{Bn} = 0.49$  eV. The transition from *n*-type to *p*-type Schottky contact can be observed at d = 2.74 Å. Our results indicate that the G/MoS<sub>2</sub> interface can be used in electronic and optoelectronic devices, such as the field effect transistors.

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